

Abstract

As growing efforts take place to enhance the operational safety of nuclear reactors, fuel composites have been explored as replacement to the traditionally used Uranium dioxide (UO_2). One potential candidate that has been gaining momentum as a fuel composite additive is Uranium diboride. UB_2 is known to have a higher uranium density and higher thermal conductivity than UO_2 , properties that would allow for a lower enrichment of the fuel pellets as well as improve the temperature gradient across the pellet during reactor operation. While various challenges arise when considering UB_2 as a drop-in replacement to UO_2 , UB_2 has shown much promise as a composite fuel when combined with other uranium compounds such as U_3Si_2 . Through the use of an arc-melter system, 50-250 mg ingots of UB_2 were fabricated using the fragments of a larger 2-5 g ingot of UB_2 . X-Ray diffraction analysis was performed to confirm the purity of the initial UB_2 ingot. Further, an infrared camera was used to monitor the temperature of the furnace chamber during the mini- UB_2 bead fabrication. The purpose of this project is to understand the fabrication process of UB_2 and characterize the micro-structure of the as-fabricated mini fuel beads. We wish to better understand the viability of UB_2 as a potential fuel composite additive.

Background

- It has been well established by the literature that Uranium dioxide has a thermal conductivity of $6.5 \text{ W m}^{-1} \text{ K}^{-1}$ at 300°C [1]. This value decreases with temperature increase which exacerbates thermal transport during an accident.
- The low thermal conductivity of UO_2 has been observed to cause a non-uniform temperature gradient across fuel pellets during reactor operation.
- This non-uniformity in temperature presents itself mainly through heat retention in the centerline of the UO_2 fuel pellets that causes fracturing and fragment dispersal.
- UB_2 has a higher thermal conductivity of $25 \text{ W m}^{-1} \text{ K}^{-1}$ at 300°C and a higher uranium density of 11.7 g-U cm^{-3} [1].
- UB_2 while a promising ATF candidate, still has its own challenges to consider.

Material Properties	UO_2	UB_2	U_3Si_2
Uranium density (g-U/cm ³)	9.6	11.7	11.34
Thermal Conductivity (W/m K at 300°C)	6.5 (95% TD)	25 (100% TD)	14.7 (98% TD)
Melting point (C)	2840	2385	1665

Fig. 1 A table showing the properties of different ATF candidates as compared to those of traditional Uranium dioxide. [1]

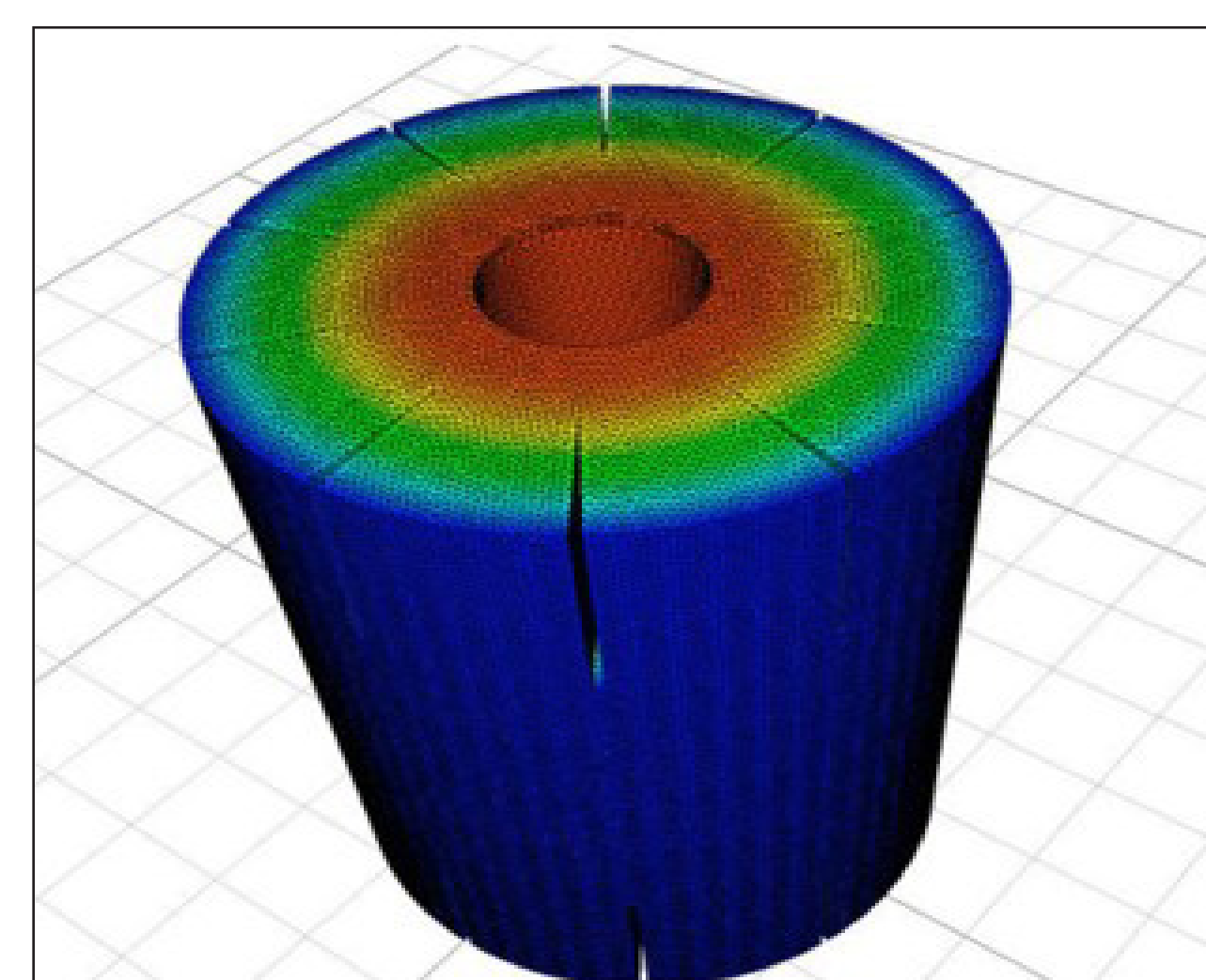


Fig. 2 An image showing the heat retention in the centerline of a UO_2 fuel pellet [2].

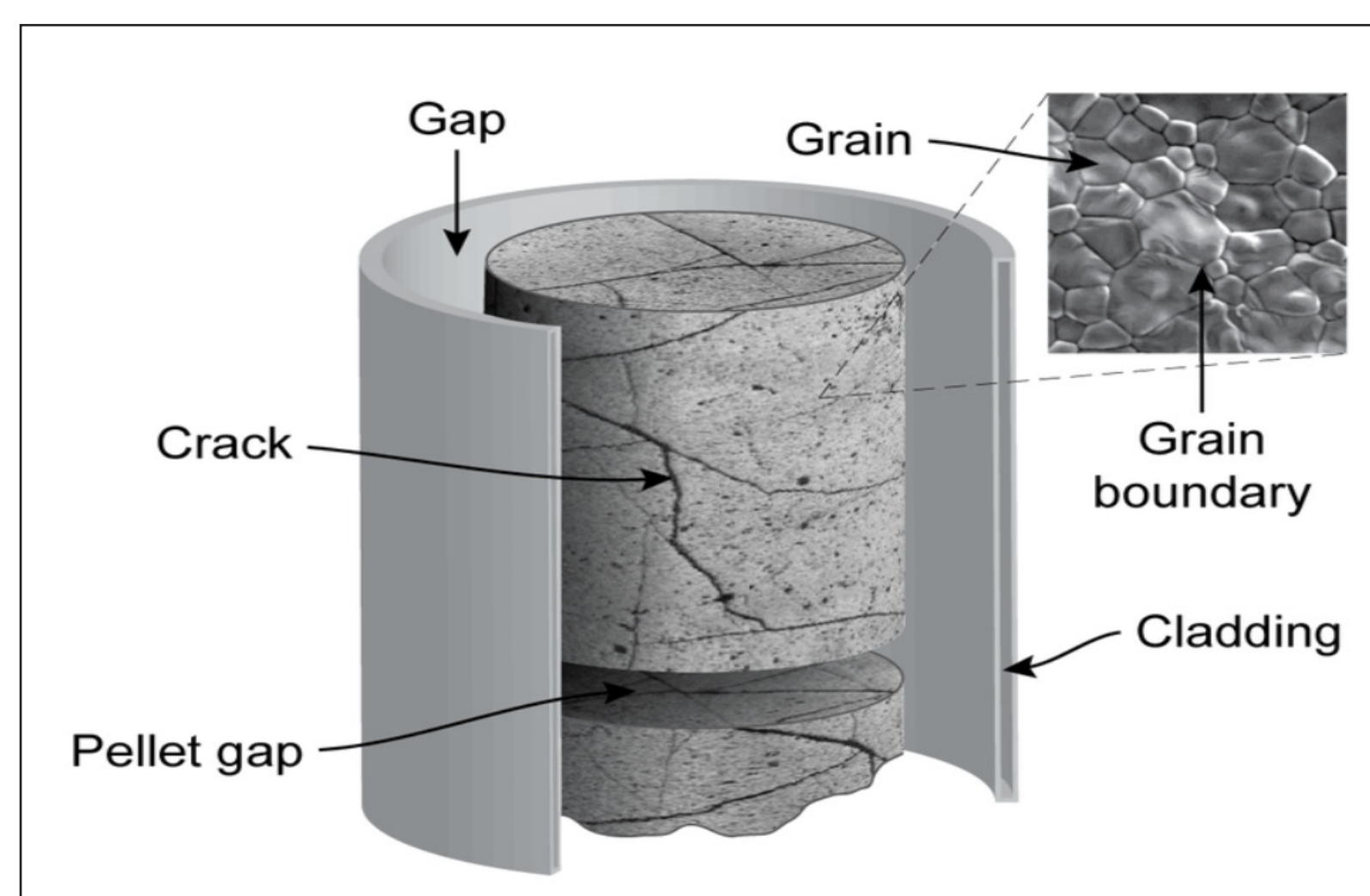


Fig. 3 An image showing the cracking of a UO_2 fuel pellet [3].

- New methods of fabrication must be investigated to allow for mass production of UB_2 .
- UB_2 has not been well investigated and its full properties have not been mapped out in their entirety. As-fabricated UB_2 via arc-melting has shown to develop impurity phases that are not desirable for reactor operation.

Methods

The initial UB_2 ingot was fabricated by consolidating a 2.5477g piece of depleted uranium and 0.2315g of elemental boron powder via a Centorr Tri-Arc melt furnace. The furnace was kept in an inert atmosphere with Ar gas. A total of 5 melts were conducted on the initial ingot to homogenize the sample. The same process was performed to remelt the large UB_2 ingot into smaller 50-250 mg ingots using a customized copper hearth. A Bruker D2 Phaser XRD was used as well as a FLIR X-Series science camera.

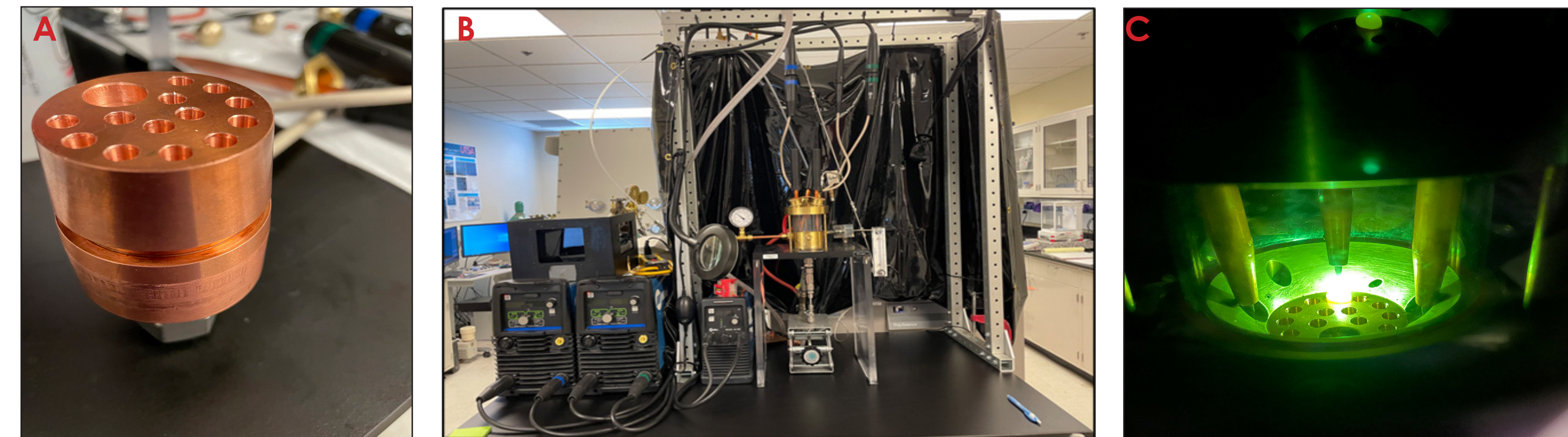


Fig. 4 A) Image of the customized copper hearth. B) Image of the Tri-Arc furnace system set-up. C) Image of the furnace chamber during operation.

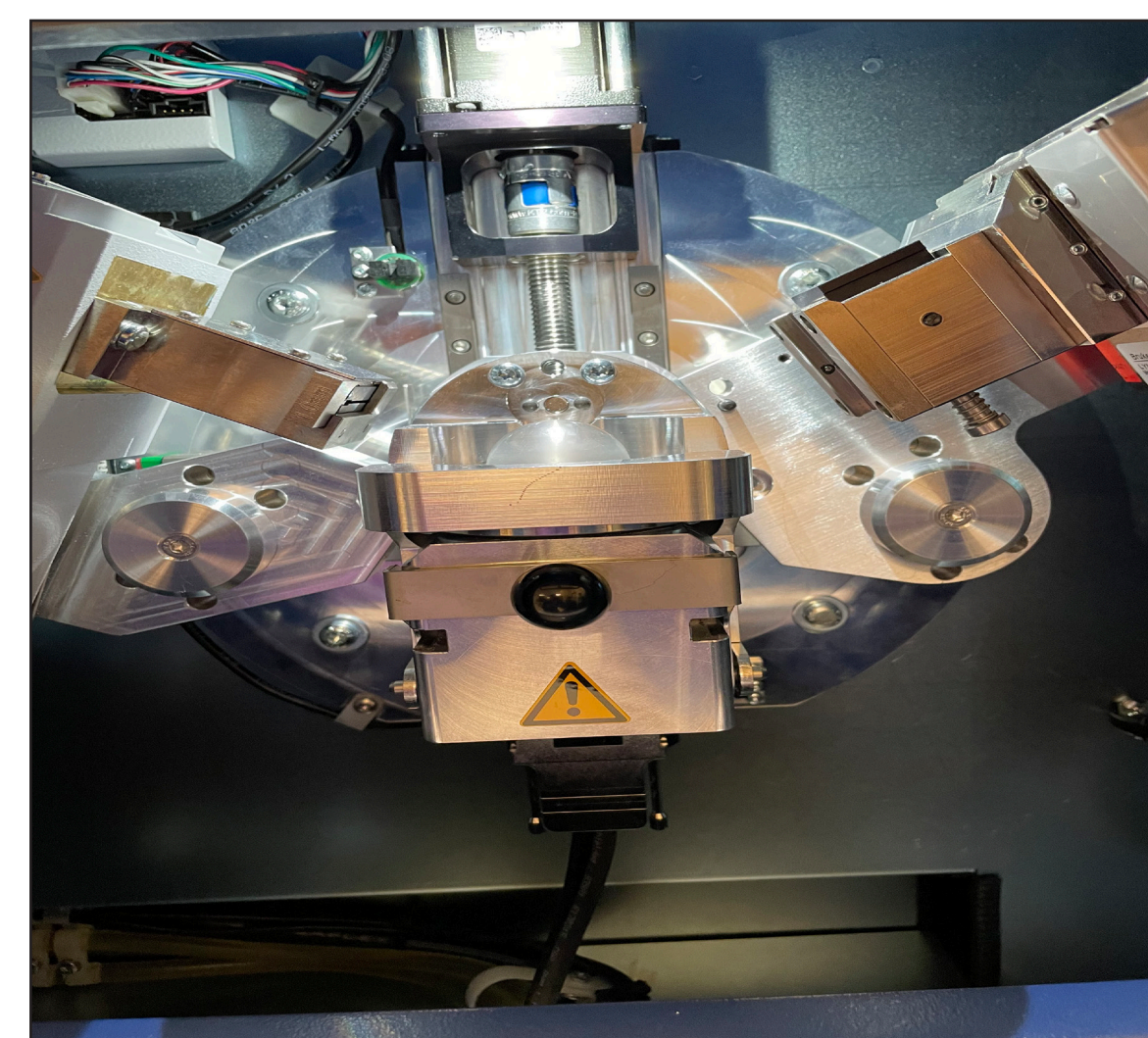


Fig. 5 An image of the interior of the Bruker D2 Phaser XRD.

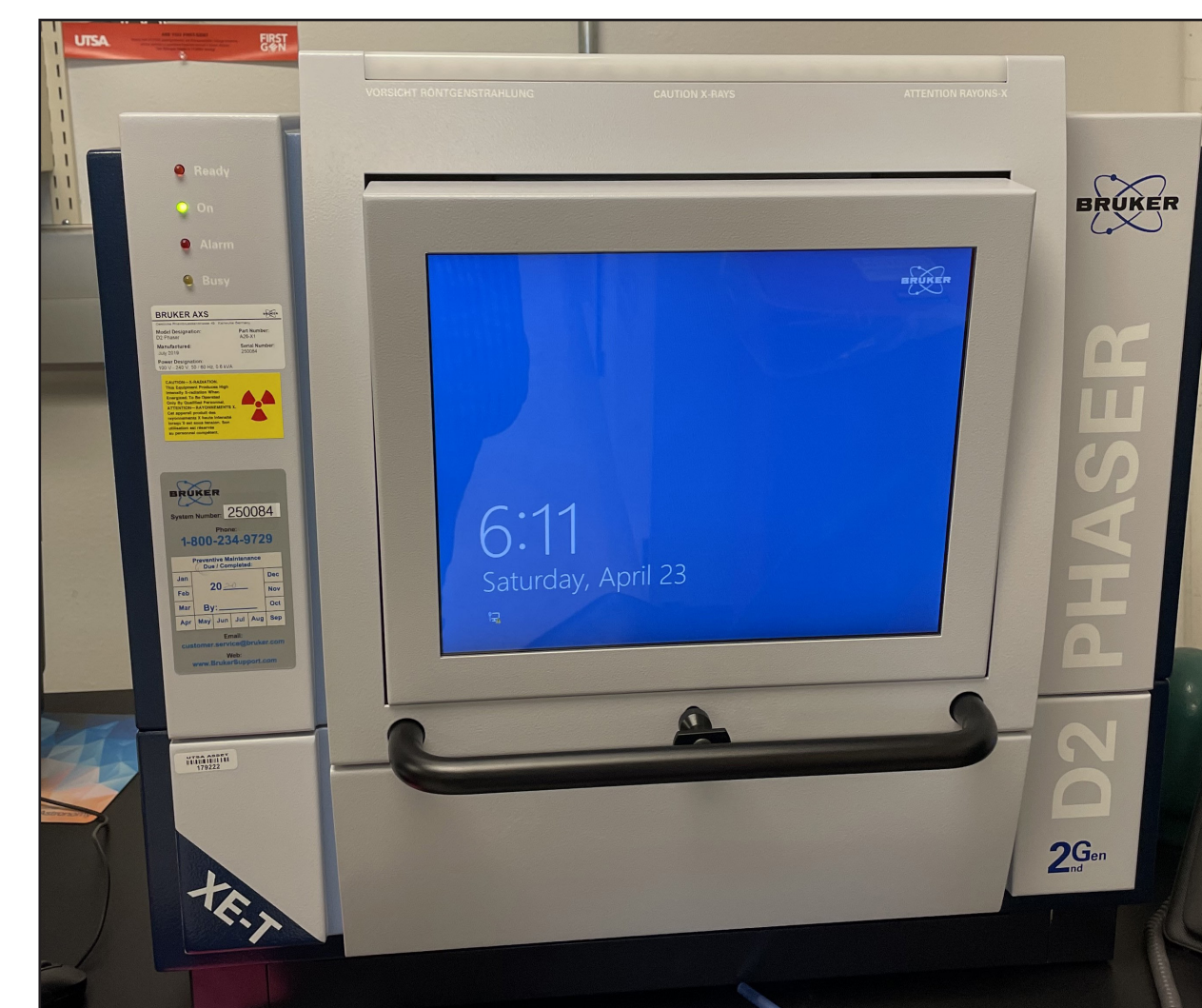


Fig. 6 An image of the exterior of the Bruker D2 Phaser XRD.

Results

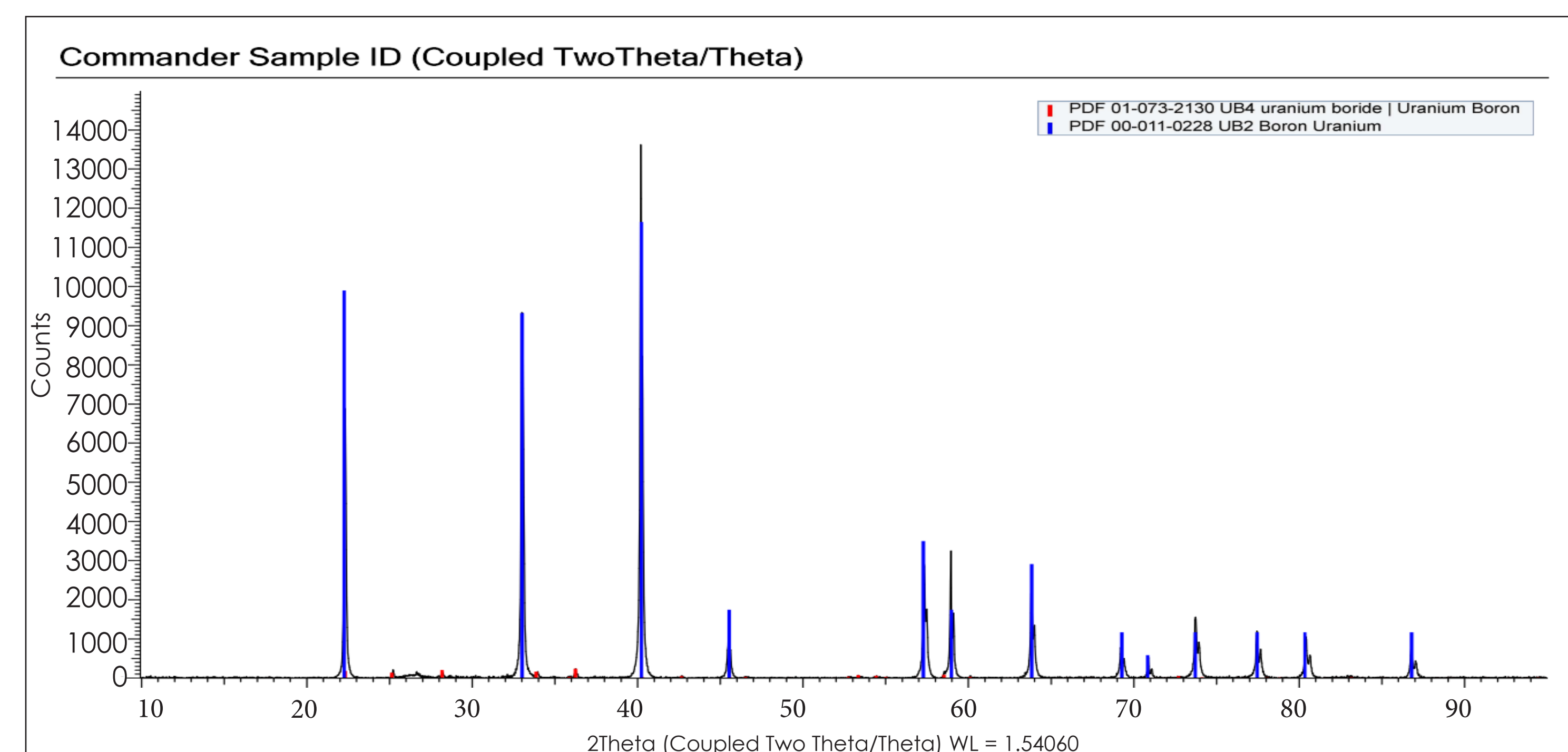


Fig. 7 Image of the initial XRD results of the large UB_2 ingot confirming a majority pure UB_2 phase

Following the fabrication of the large UB_2 ingot, X-ray diffraction (XRD) analysis was performed to confirm the purity of the sample. The bench-top Bruker D2 Phaser XRD was run from 10 to $95^\circ 2\theta$ with a 0.1 step and dwell time of 1 second. XRD results confirmed the dominant presence of UB_2 and low trace amounts of UB_4 ; these results are outlined in figures 7 and 8. A FLIR X-series infrared camera was used during the fabrication of the mini- UB_2 beads to monitor the temperature inside the furnace chamber.

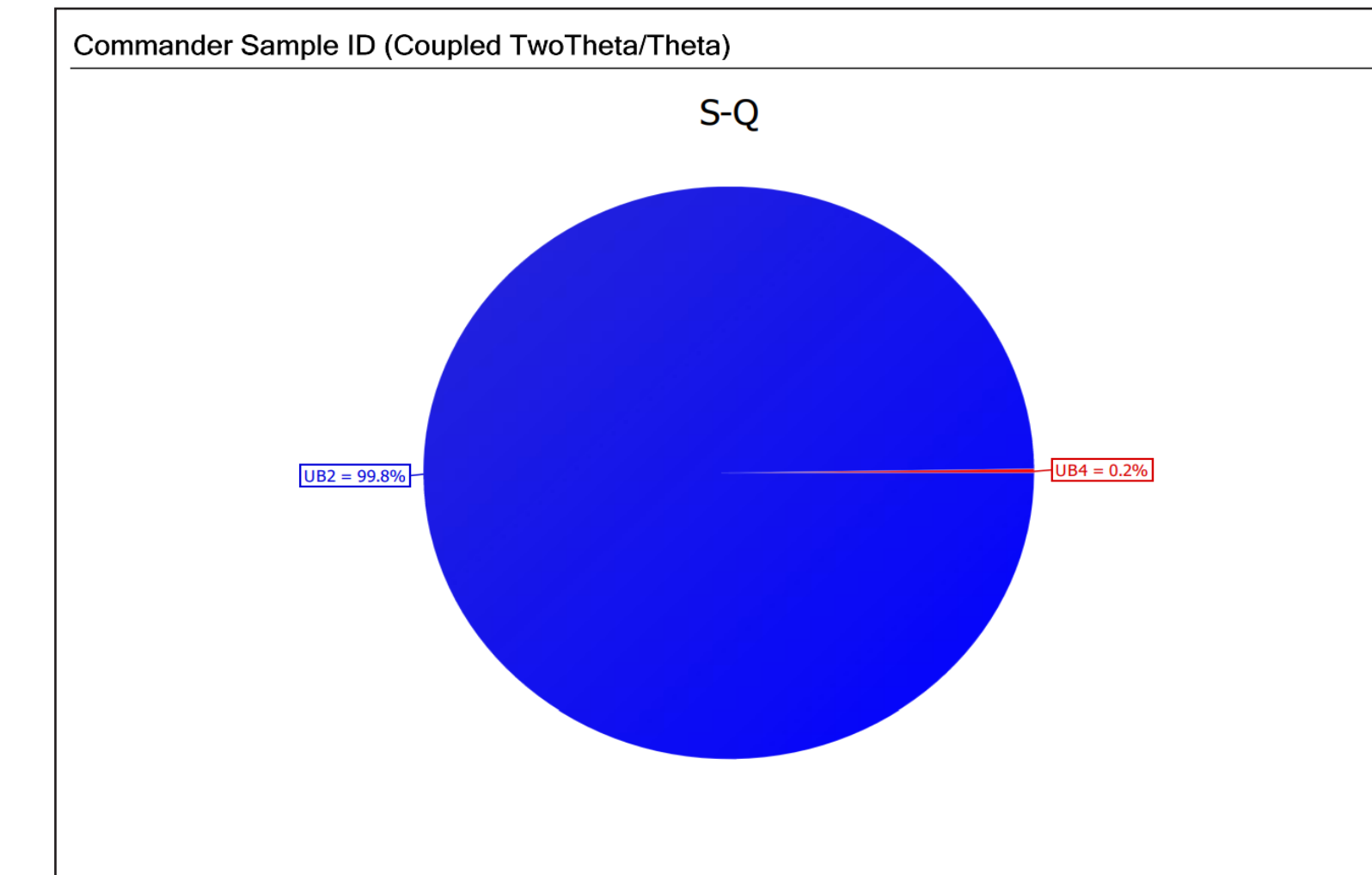


Fig. 8 Image of the initial XRD results of the large UB_2 ingot confirming a majority pure UB_2 phase and trace amounts of UB_4

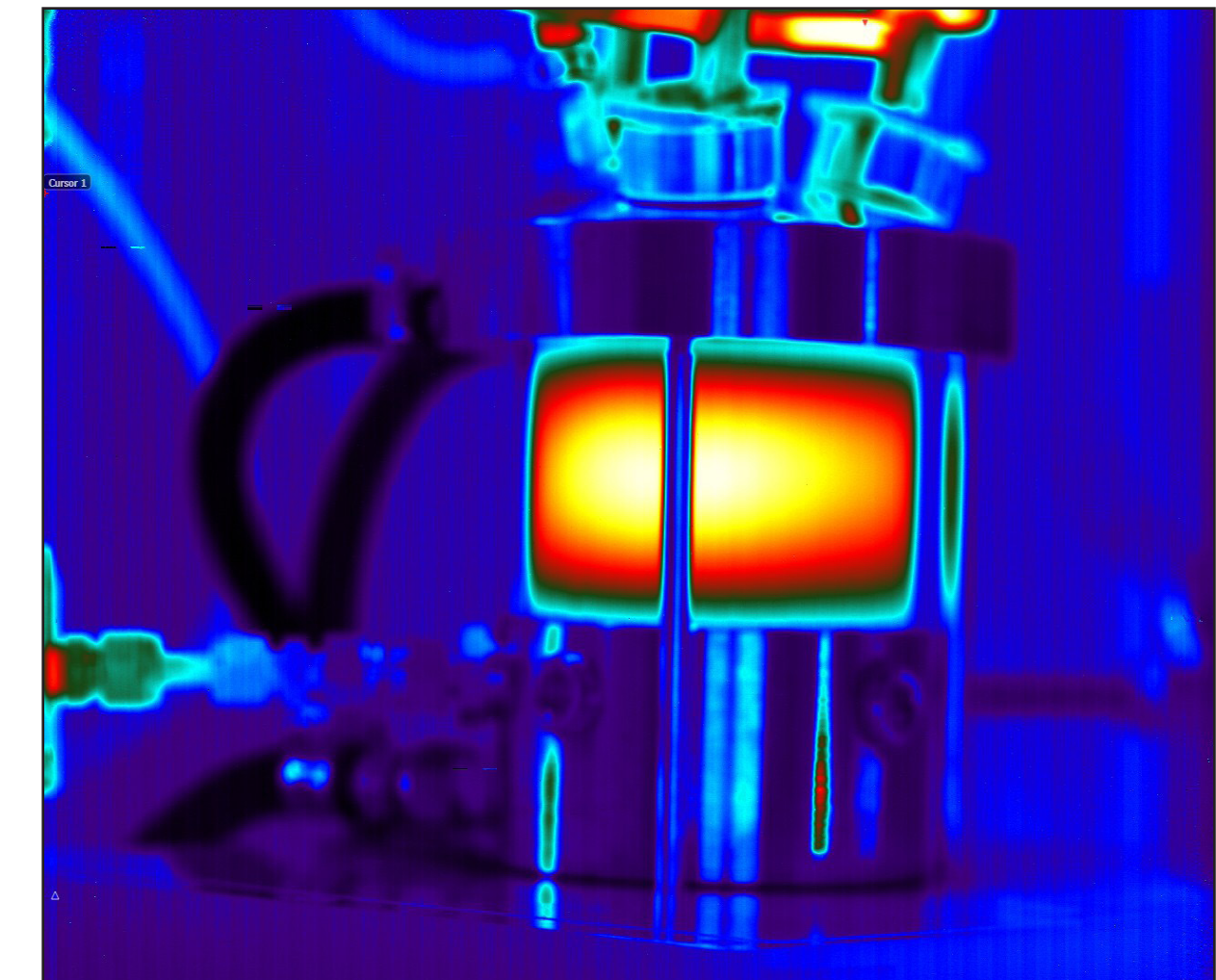


Fig. 9 IR image of the furnace chamber during mini- UB_2 bead fabrication.

Unfortunately, the IR camera lens used was not calibrated for temperatures above 350°C resulting in inaccurate temperature readings. Despite the incorrect calibration, using control IR images, we approximated the temperature inside the furnace chamber during the melt to reach $\sim 1900^\circ \text{C}$ at the highest amperage setting used on the stingers.

Conclusion

- Fabrication of the mini- UB_2 beads via arc-melting proved satisfactory as sufficient UB_2 purity was confirmed by XRD analysis.
- The temperature at which the melt was performed is still unclear as the IR camera used to monitor it was not correctly calibrated.
- Further monitoring and investigation of the temperatures within the furnace chambers is needed to inform on melt performance.
- To better understand the viability of UB_2 as a fuel additive, oxidation testing is needed to inform on its performance under off-normal reactor conditions.
- Characterization of the as-fabricated as well as heat treated UB_2 sample is needed to understand the different phases present. This is to be carried out via Scanning Electron Microscopy.
- Overall, arc-melting was proved a reliable method of producing UB_2 as confirmed by XRD analysis. As arc-melting is not a effective method for mass producing UB_2 , additional study is needed to look into more efficient methods to accomplish this.

References

- Jennifer K. Watkins, Adrian R. Wagner, Adrian Gonzales, Brian J. Jaques, Elizabeth S. Sooby, Challenges and opportunities to alloyed and composite fuel architectures to mitigate high uranium density fuel oxidation: Uranium diboride and uranium carbide
- Piro, Markus (2011) et al, Computation of Thermodynamic Equilibria Pertinent to Nuclear Materials in Multi-Physics Codes.
- Fors, Patrik. (2009). The effect of dissolved hydrogen on spent nuclear fuel corrosion.. 10.13140/RG.2.1.5125.4244.

Acknowledgements

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